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# Changes in electronic structure by Li ion deintercalation in $\text{LiNiO}_2$ from nickel $L$ -edge and O $K$ -edge XANES

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## Abstract

Change in electronic structure by lithium ion deintercalation in  $\text{LiNiO}_2$  was investigated by Ni  $L$ -edge and O  $K$ -edge X-ray absorption near edge structure (XANES). The Ni  $L$ -edge XANES for  $\text{LiNiO}_2$  indicated the Ni occurs in  $\text{LiNiO}_2$  as  $\text{Ni}^{2+}$  ions. The Ni  $L$ -edge XANES for  $\text{Li}_x\text{NiO}_2$  showed no chemical shift to indicate that the Ni ion in  $\text{Li}_x\text{NiO}_2$  is still  $\text{Ni}^{2+}$ , even at low  $x$  value. The O  $K$ -edge XANES for  $\text{Li}_x\text{NiO}_2$  also indicated that the holes compensating the lithium ion deintercalation are located primarily in oxygen 2p states rather than in Ni 3d states. © 2001 Elsevier Science B.V. All rights reserved.

*Keywords:* Nickel  $L$ -edge; O  $K$ -edge;  $\text{LiNiO}_2$

## 1. Introduction

The  $\text{LiNiO}_2$  oxide with a layered  $\alpha$ - $\text{NaFeO}_2$  structure is one of the most promising cathode materials used in 4 V-type lithium ion batteries because its low cost and high theoretical energy density [1–3]. It is important to clarify the change of electronic structure during charge and discharge process in order to understand the electrochemical properties. Recently, first-principle molecular orbital calculations of the electronic structure of the  $\text{Li}_{1-x}\text{NiO}_2$  have indicated that lithium ion deintercalation increases the covalent interaction between Ni and oxygen, and that the oxidation associated with the deintercalation mainly takes place on oxygen [4–6]. However, experimental studies of the electronic structure change during the lithium ion deintercalation are insufficient.

In this study, Ni  $L_{2,3}$ -edge and O  $K$ -edge X-ray absorption near edge structure (XANES) was determined the electronic structure of  $\text{LiNiO}_2$  during lithium ion deintercalation.

## 2. Experimental

The  $\text{LiNiO}_2$  powder was prepared by conventional solid state reaction starting with lithium hydroxide (Wako Chemical Co. Inc., 99.9%) and nickel hydroxide (Wako Chemical Co. Inc., 99.9%). A mixture of  $\text{Li}(\text{OH})$  and  $\text{Ni}(\text{OH})_2$  in a mole ratio of 1:1 was heated at  $770^\circ\text{C}$  for 30 h in an oxygen atmosphere. The crystal structure of the product was determined by XRD using Mo  $K\alpha$  radiation. The  $\text{Li}_x\text{NiO}_2$  was prepared by electrochemical lithium deintercalation. A mixture of 82.5 wt.%  $\text{LiNiO}_2$ , 15 wt.% acetylene black, and 2.5 wt.% polytetrafluoropropylene binder was used as working electrode. The electrolyte was a 1 M  $\text{LiClO}_4$  in PC solution.

The Ni  $L_{2,3}$ -edge XANES spectra were measured on the BL-7A beam line and O  $K$ -edge XANES were measured on the BL-8B1 beam line at UVSOR (Okazaki, Japan) with a ring energy of 750 MeV in a mode of total electron yield at room temperature.

## 3. Results and discussion

The XRD pattern for  $\text{LiNiO}_2$  was indexed in a rhombohedral  $R\bar{3}m$  space group [1]. The first-cycle discharge capacity is  $140 \text{ mAh g}^{-1}$ .

Fig. 1 shows the Ni  $L$ -edge XANES of  $\text{Li}_x\text{NiO}_2$  for various  $x$  values. The Ni  $L$ -edge XANES shows two strong

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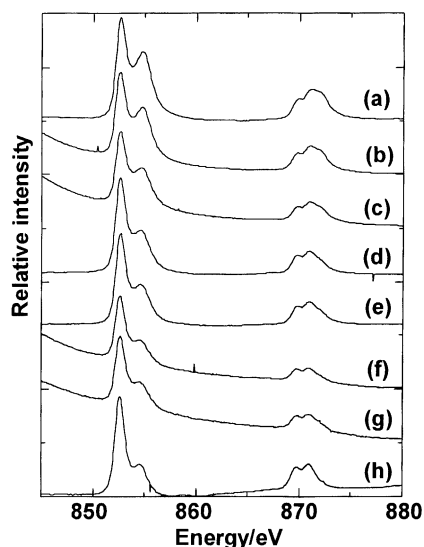


Fig. 1. Ni 2p absorption spectra for  $\text{Li}_x\text{NiO}_2$ : (a) 0.2, (b) 0.4, (c) 0.5, (d) 0.6, (e) 0.7, (f) 0.8, (g) 0.9, (h) 1.0.

absorption features of the spin-orbit splitting of the Ni 2p core hole [7]. The absorption near 852–856 eV is the  $2p_{3/2}$  (L3) edge and that near 868–872 eV is the  $2p_{1/2}$  (L2) edge. The Ni *L*-edge XANES for  $\text{LiNiO}_2$  is in good agreement with that reported by Montoro et al. [8]. The XANES for  $\text{LiNiO}_2$  is similar to that for NiO, which shows the Ni ions in  $\text{LiNiO}_2$  to be  $\text{Ni}^{2+}$  ions in a high-spin state [7,8]. The spectra for  $\text{Li}_x\text{NiO}_2$  ( $x = 0.2, 0.4, 0.5, 0.6, 0.7, 0.8, 0.9, 1.0$ ) do not exhibit chemical shift and the change in shape are small, indicating the Ni ion in the  $\text{Li}_x\text{NiO}_2$  is still  $\text{Ni}^{2+}$  even at as low an  $x$  value as 0.2.

Fig. 2 shows the O *K*-edge XANES for  $\text{Li}_x\text{NiO}_2$  for various  $x$  values. The peak at about 532 eV (●) is attributed to oxygen in the substrate. A peak at about 533 eV is attributed to the band derived from the mixing of the Ni 3d states with O 2p states. The broad structure near 535–550 eV is attributed to a Ni band with 4sp character. The peak at near 528 eV increased with decreasing lithium

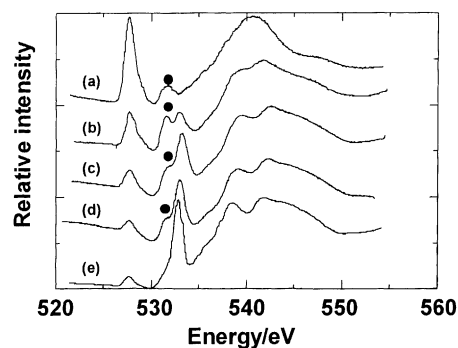


Fig. 2. O 1s absorption spectra for  $\text{Li}_x\text{NiO}_2$ : (a) 0.2, (b) 0.5, (c) 0.6, (d) 0.8, (e) 1.0.

content. This result shows that oxidation influences the oxygen 2p orbital and the ground state of  $\text{Li}_x\text{NiO}_2$  ( $x < 1.0$ ) is  $\text{Ni}^{2+}\text{L}$ , where L represents a ligand hole state [9]. This indicates that the lithium deintercalation reaction does not involve Ni but oxygen. This is in good agreement with the result of first-principles calculations for the  $\text{Li}_x\text{NiO}_2$  system [4–6].

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